



A study of size-controlled Au@Cu₂O nanocomposite for highly improved methyl orange catalytic performances

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ABSTRACT

In this study, we synthesized Au-Cu₂O core-shell nanoparticles (Au@Cu₂O NPs) with epitaxial Cu₂O shell, which have highly improved catalytic performances in methyl orange reduction reaction. We have obtained Au@Cu₂O NPs with different sizes at low temperatures by changing the amount of Cu²⁺ precursor. Both Au and Cu₂O are catalytic materials, especially Cu₂O, as a *p*-type semiconductor, whose catalytic ability depends on the crystalline facets. Catalytic reduction of methyl orange (MO) was used as a model system to explore the catalytic properties of Au@Cu₂O nanocomposite. In comparison to pure Au and Cu₂O NPs, the catalytic performance of Au@Cu₂O has a noticeable improvement. The best catalytic rate was ~ 22 times faster than that of AuNRs and ~ 4 times than that of Cu₂O NPs. By studying the catalytic mechanism, it is supposed that the Schottky barrier at the Au-Cu₂O interfaces leads to the charge separation, which is beneficial to catalysis. Therefore, the Au@Cu₂O NPs we designed with controllable shell thickness is an ideal composite material in the catalyst domain.

Introduction

At present, the property of single material has been incapable to meet the demand of actual applications, therefore hybrid nanostructures have obtained more attention from all kinds of fields. Usually, the composite materials not only keep their original functions

of individual component but also have other synergistic effects, producing novel properties in the aspect of physics, chemistry, and bio-medicine [1–5]. What's more, compared to nanostructures with a single material, multi-component nanostructures are easier to improve their properties by changing the size, proportion, morphology, and geometry of

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different components [6–8]. Metal–semiconductor nanostructures have generated considerable research interests among the multi-component nanostructures, because the combination of metal and metal oxides induces electron enrichment phenomenon, which benefits charge separation [9, 10]. In addition, semiconductor materials have very low cost. Metal–semiconductor nanostructures can reduce production costs while they achieve the improved performances, in comparison to other hybrid nanostructures.

Decorating metal nanoparticles or supported by a novel metal core onto semiconductor materials, metal–semiconductor materials can be used in biological detection, catalysis, and other applications [11–14]. Au nanostructures have been widely used in catalysis, especially Au nanorods (AuNRs), having an excellent localized surface plasmon resonances (LSPRs) property, and their LSPRs can be precisely controlled by the aspect ratio (AR) [15–18]. In addition, Cu_2O , a p-type semiconductor, having distinguished optical, electronic, and catalytic properties [19–21], is similarly attracting more focus. Hence, numerous current experiments that study the morphology and properties of Au– Cu_2O hybrid nanostructures have been carried out [22–29].

In terms of property, the synergistic effect of Au– Cu_2O nanoparticles is influenced by the structure, morphology, size or growth mode of Au and Cu_2O . Lu's group [30] have designed gold nanospheres with hollow Cu_2O shell to control their optical properties, but the thin Cu_2O shell is irregular and incomplete. In addition, Au nanospheres as a core, which have the isotropic shape, cannot offer rich plasmonic modes. AuNRs covered by Cu_2O octahedra shell consisting of {111} facets with accurate size control have been synthesized by Kong et al. [31] to enhance the catalytic ability, but with the combination of the spectra and catalytic experiments, the physical properties of AuNRs cannot be fully utilized owing to the thick Cu_2O shell. Therefore, there is still plenty of scope for optimization of the morphology of Cu_2O shell. Xu's group [32] grew a series of Au– Cu_2O hybrid structures to study the internal energy transfer of the hybrids by generating the photocurrent, and found that the core–shell structures had better Schottky effect, which is benefit for electron transfer and can be used in the catalysis field.

In this paper, we obtained an Au– Cu_2O core–shell nanocomposite using large AuNRs to overgrow the thickness-suitable and even Cu_2O shell, which can

supply a bigger contact area and more catalytic active sites. To improve the Cu_2O coverage rate, SDS and CTAB were used as binary surfactants. We also slowed down the reaction rate by changing the temperature of the growth solution from room temperature to 5 °C. And we controlled the Cu_2O shell thickness by changing the volume of Cu^{2+} precursor. Finally, by reducing methyl orange (MO) with the assistance of NaBH_4 , we explored the catalytic performance of Au@ Cu_2O nanostructure related to the thickness of Cu_2O and studied the corresponding catalytic mechanism.

Experimental

Chemicals

Hexadecyltrimethylammonium bromide (CTAB, > 99%) was purchased from TCI America. L-ascorbic acid (AA, > 99.99%), chloroauric acid (HAuCl_4), hydrochloric acid (HCl, 37 wt. % in water), copper (II) chloride dihydrate ($\text{CuCl}_2 \cdot 2\text{H}_2\text{O}$, 99.99%), sodium dodecyl sulfate (SDS, > 92.5%), hydroxylamine hydrochloride ($\text{NH}_2\text{OH} \cdot \text{HCl}$, 99.99%) and methyl orange (MO, 96%) were purchased from Shanghai Macklin Biochemical Technology Co., Ltd. Sodium borohydride (NaBH_4 , 98%), sodium oleate (NaOL , > 99.8%), silver nitrate (AgNO_3 , > 99.8%) were purchased from Shanghai Aladdin Biochemical Technology Co., Ltd. Sodium hydroxide (NaOH , > 96.0%), Ethanol (EtOH , > 99.7%) were purchased from Hangzhou Gaojing Fine Chemical Industry Co., Ltd. Ultrapure water (Millipore Milli-Q grade) with a resistivity of 18.2 M Ω was used in all of the experiments. All chemicals were used as received without further purification.

Synthesis of AuNRs

The AuNRs were synthesized using a previously reported “seed-mediated method” [33] with slightly modification. For the seed solution, 0.25 mL of 10.0 mM HAuCl_4 and 10 mL of 0.1 M CTAB were mixed in a 20 mL bottle. The freshly prepared ice-cold NaBH_4 (0.6 mL, 10.0 mM) was quickly added to the mixed solution under the vigorous agitation. After the solution became brownish yellow, the stirring was stopped and let the solution stand for about 30 min before use.

For the growth solution, 7.0 g of CTAB and 1.24 g of NaOL were added in a 500 mL flask, followed by the addition of 250 mL water, magnetically stirred and heated in 50 °C water bath until it's completely dissolved. The solution was cooled down to 30 °C, after which 24 mL of 4.0 mM AgNO_3 was added and kept it for 15 min. 25 mL of 10.0 mM HAuCl_4 and 225 mL of water were added and the solution was magnetically stirred at 700 rpm for 1 h until the color of solution faded. 1.5 mL of HCl (37 wt.% in water) was added under the stirring at 700 rpm for 15 min. After that, 1.25 mL of 64.0 mM AA was added and the mixture was vigorously stirred for 30 s. Finally, 0.4 mL of seed was injected into the mixture with stirring for 30 s. The subsequent rod growth was performed under static conditions overnight to obtain AuNRs. The resulting AuNRs have a LSPR wavelength of 752 nm. The final AuNRs was centrifuged at 8000 rpm for 15 min, washed once with water, and redispersed in 50 mL of 1.5 mM CTAB.

Synthesis of Au@Cu₂O NPs

The prepared AuNRs, dispersed in 1.5 mM CTAB, have a maximum of absorption peak about 1.35 when it is diluted 16.7 times in a 1 cm cuvette. Y μL of 0.05 M $\text{CuCl}_2 \cdot 2\text{H}_2\text{O}$ aqueous solution were added in X mL of 1.5 mM CTAB and followed by adding 0.5 mL of 0.3 M SDS. 0.3 mL of AuNRs solution was added and then the mixed solution was magnetically stirred (300 rpm, 10 min). The solution was kept static at 5 °C for 5 min. 65 μL of 0.1 M NaOH were added, followed by adding 175 μL of 0.1 M $\text{NH}_2\text{OH} \cdot \text{HCl}$. The solutions were mixed and stirred for 10 s. The mixture left undisturbed at 5 °C for 2 h. Then, the resulting solution was centrifuged at 4000 rpm for 5 min to remove the surfactant. The product was washed with deionized water once and ethanol twice, and stored in 1.5 mL ethanol. The thickness of the Cu₂O shell was controlled by changing the volume of the CuCl_2 solution. The value of y we chosen was 35 μL , 65 μL , and 100 μL , and the Au@Cu₂O NPs synthesized with three sizes was termed as AC-35, AC-65, and AC-100, respectively. X was adjusted to ensure 5 mL of the total volume.

Synthesis of Cu₂O NPs

The Cu₂O NPs were obtained by a method similar to getting Au@Cu₂O NPs. 65 μL of 0.05 M $\text{CuCl}_2 \cdot 2\text{H}_2\text{O}$

aqueous solution was added in 3.9 mL of water, followed by adding 0.5 mL of 0.3 M SDS. 0.3 mL of AuNRs solution was added with magnetically stirring at 300 rpm for 10 min. Let it undisturbed at 5 °C for 5 min. Afterward, 65 μL of 0.1 M NaOH was added, followed by adding 175 μL of 0.1 M $\text{NH}_2\text{OH} \cdot \text{HCl}$. The solutions were mixed and stirred for 10 s. The mixture was aged at 5 °C for 2 h. Then, the product was obtained by centrifugation and washed with deionized water once and absolute ethanol twice, and finally stored in 1.5 mL ethanol.

Catalytic reduction of MO

The prepared catalytic nanoparticles (AC-35/65/100, AuNRs, Cu₂O NPs) were centrifuged and uniformly dispersed in Milli-Q water by ultrasonication. In a cuvette, 1 mL of MO (0.1 mM) and 16 μL of catalyst were added, followed by the addition of a freshly prepared NaBH_4 (50 μL , 0.05 M) solution. After adding the catalyst, the solution gradually changed from orange to colorless, which was monitored by measuring the change in absorbance of MO at 464 nm by UV–vis spectra. The spectra were recorded for a regular interval time of 1 min. And each reaction has been repeated three times to minimize the test error.

FDTD calculations

The extinction spectra simulations of the Au@Cu₂O nanostructures were calculated using the finite-difference-time-domain (FDTD) method with a software package from FDTD Solutions (Lumerical Solutions, Inc.). The refractive index of the surrounding medium was taken to be the refractive index of water (1.33). The boundary conditions of the simulation are perfectly matched layer (PML). The boundary with NPs inside was divided into mesh size of 2 nm. A plane wave from 300 to 1200 nm wavelength was illuminated. The Au core in the Au@Cu₂O NPs was modeled as a cylinder capped with two half spheres at both ends. And the Cu₂O shell was modeled as a cuboid with rounded corners and edges. For the longitudinal excitation, the incident light was perpendicular to the vertical axis and polarized along the long axis of gold nanorods. For the transverse excitation, both the incident light and polarization direction were perpendicular to the long axis. Dielectric functions for Au and Cu₂O were provided by Palik [34]. To investigate the influence of

anisotropy on the extinction spectra, we also simulated the Au@Cu₂O with AR of 1, 2, 3, and 4, respectively.

Characterizations

Optical extinction spectra were recorded with a UV-1900i spectrophotometer (SHIMADZU, Japan) with a 1.0 cm optical path. The transmission electron microscope (TEM) images were obtained with a HT-7700 microscope (HITACHI, Japan) operated at 100 kV. High resolution TEM (HRTEM) images and energy disperse spectroscopy (EDS) mapping were performed by a Fei-Talos-F200S operated at 200 kV. The particle sizes of the nanoparticles were measured from TEM images, whereby > 50 nanoparticles were measured for each sample.

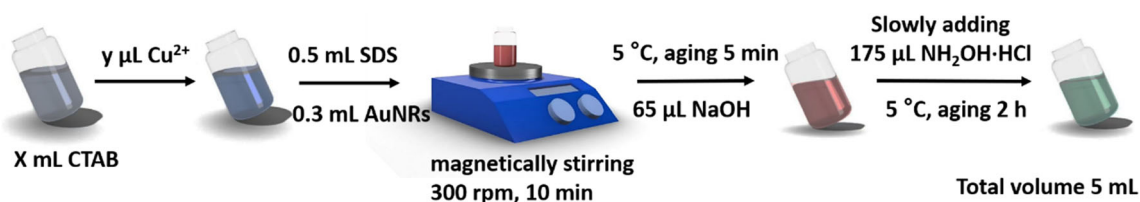
Results and discussion

The AuNRs used for further synthesis of Au@Cu₂O NPs were obtained by seed-mediated method with binary surfactant mixtures [33]. Their extinction spectra and morphology characterization are given in Figure S1 in the Supporting Information. The AuNRs sample with good monodispersity have an average length of 112 ± 9 nm and the AR of ~ 3 , which have a narrow longitudinal peak at 752 nm. The synthetic process of Au@Cu₂O NPs was shown in Scheme 1. In the CTAB solution with CuCl₂·2H₂O as the source of copper, SDS surfactant, concentrated AuNRs solution, NaOH and NH₂OH·HCl were orderly added. NH₂OH·HCl was used as the reducing agent. When NH₂OH·HCl was added, small Cu₂O seed particles were formed. These seed particles quickly aggregated into larger particles and attached to the AuNRs [35]. After aging for 2 h, larger Cu₂O particles grow into the dense shell and the final product was obtained.

To cover a uniform Cu₂O shell, CTAB and SDS were used as binary surfactant mixtures, and the

mixture solution needed to age in a refrigerator at 5 °C before adding NH₂OH·HCl. The reason for choosing binary surfactant is that AuNRs are easy to aggregate during the process of magnetically stirring using SDS only, or at a low concentration of CTAB. Figure S2 show the results of using CTAB < 1 mM, indicating that the aggregated AuNRs cannot coat with intact Cu₂O shells on their surface. Lower temperature is designed to slow down the growth speed of Cu₂O shell, because Cu²⁺ was rapidly reduced to Cu₂O at room temperature, which is unfavorable for uniform coverage. At the lower temperature condition (5 °C), the change rate of solution color is apparently slow, and the final product has good monodispersity and uniform coverage.

The high-resolution transmission electron microscopy (HRTEM) image of a single Au@Cu₂O particle is shown in Fig. 1a. Obvious bright/dark contrast indicates that the Au nanorod locates in the center of the cubic particle. Figure 1b shows the enlarged picture of the red square region of Fig. 1a, which is the interface between Au core and Cu₂O shell on the body part of Au@Cu₂O, revealing oriented epitaxial growth and the single-crystal nature of Cu₂O shell. Cu₂O (200) is parallel to Au (200). The same TEM analysis was performed in Figure S3b, which chose the tips of the Au@Cu₂O nanoparticle in red square region of Figure S3a. Similarly, the {200} facets of Cu₂O grown along the {200} facets of Au. The FFT of Cu₂O shell in Fig. 1c reveals the Cu₂O shells are mainly constructed by {200} facets and {110} facets, which is owing to a larger concentration of NH₂OH·HCl added. The amount of NH₂OH·HCl may influence the growth rate of different facets [35], which have a crucial effect on the catalytic property. High-angle annular dark-field scanning transmission electron microscopy (HAADF-STEM) images of Au@Cu₂O nanoparticles are shown in Figure S3c. A core-shell structure that a light Cu₂O shell around a bright Au core was obviously exhibited. The elemental distribution of Au, Cu, and O from the



Scheme 1 Schematic illustration of the synthesis of Au@Cu₂O NPs.

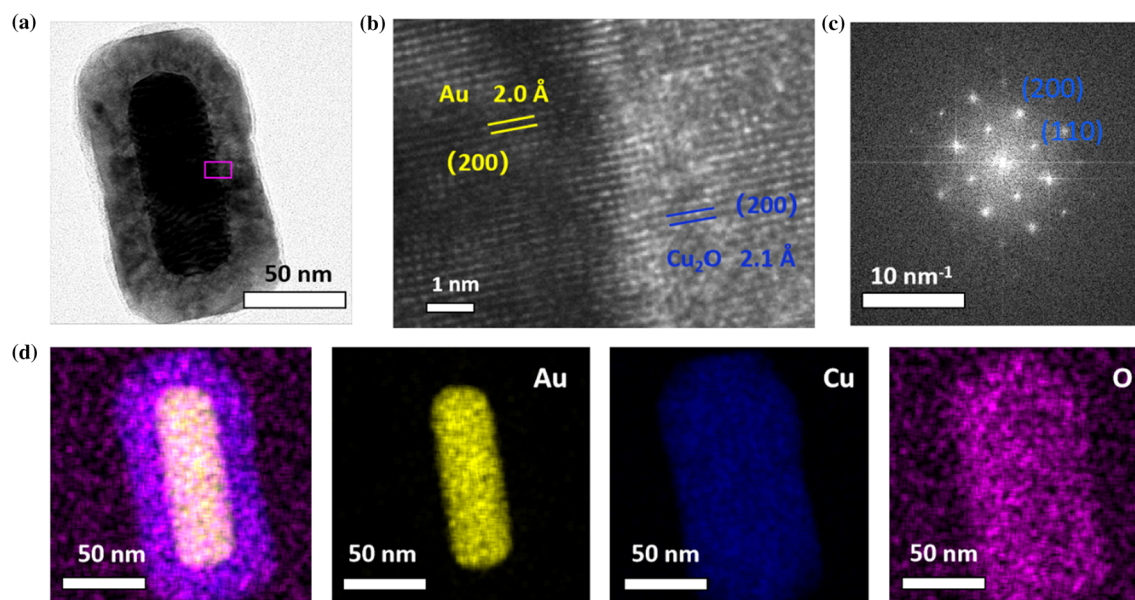


Figure 1 Characterization of the Au@Cu₂O NPs. **a** HRTEM images of a single AC-65 particle (the Au@Cu₂O with Cu₂O shell of 21 nm). **b** Interfacial HRTEM images of AC-65 sides of the red rectangular region in **a**, the (200) planes of Cu₂O were found to

grow epitaxially on the (200) planes of the AuNR. **c** The corresponding FFT in marked region of Cu₂O shell shown in panel **a**. **d** EDS elemental mapping images: spatial distribution of Au, Cu and O, respectively.

energy-dispersive spectroscopy (EDS) mapping in Fig. 1d was taken to further confirm the core-shell Au@Cu₂O nanostructure.

The method that NH₂OH·HCl as a mild reducing agent reduces Cu²⁺ with binary surfactant mixtures at lower temperatures can easily grow dense and uniform Cu₂O shells over the surface of AuNRs with good coverage. Therefore, we can control the shell thickness of Cu₂O by changing the volume of Cu²⁺ added at a fixed concentration and the volume of AuNRs. A series of 0.05 M Cu²⁺ were added, namely, 35 μ L, 65 μ L, 100 μ L, and the obtained Au@Cu₂O NPs were termed as AC-35, AC-65, and AC-100, which had increased shell thicknesses. The morphological detail of the single particle was illustrated in TEM images shown in the upper-right inset of Figs. 2a–c. Their thicknesses of Cu₂O are about 15 nm, 21 nm, and 30 nm, respectively. As shown in TEM images with a number of sample particles, each nanoparticle contained only one Au nanorod inside, and every nanostructure had very uniform shell, verifying the approach has a high productivity of monodispersed Au@Cu₂O NPs.

The relation between the optical properties of Au@Cu₂O NRs and the shell thickness was further explored by taking the UV–vis absorption spectra of AuNRs and Au@Cu₂O NRs with different shell

thicknesses. The extinction spectra of AuNRs, AC-35, AC-65, and AC-100 were shown in Fig. 3a, which were measured at the same concentration. Compared with the bare AuNRs, the extinction spectra of Au@Cu₂O NRs are abundant but intensity has an apparent decrease, which means covering the Cu₂O shell over the AuNRs surface weakened surface plasmon resonance, especially the longitudinal plasmonic peak. With the increase of the Cu₂O shell thickness, this longitudinal peak is more and more apparent and has a slight red-shift, owing to a large permittivity of Cu₂O ($\epsilon \sim 7$) [36]. Au@Cu₂O crystals have a characteristic absorption peak at ~ 500 nm arising from the scattering of the Cu₂O shell [37]. It's worth noting that two transverse plasmonic peaks, signed as peak1 and peak2 in Fig. 3a, were observed, which originated from an octupole plasmon resonance mode and the transverse dipole plasmon mode, respectively [37]. Two transverse plasmon peaks produces maybe due to the component, the AR, or existing gap and so on [38]. In this structure, we infer that the Cu₂O shell with different thicknesses is the major factor. The intensity and wavelength of two peaks were extracted and drawn as a function of Cu²⁺ amount in a line chart of Figs. 3b and c. The red-shift trend of peak1 and peak2 is same as the longitudinal plasmon peak. It is interesting that

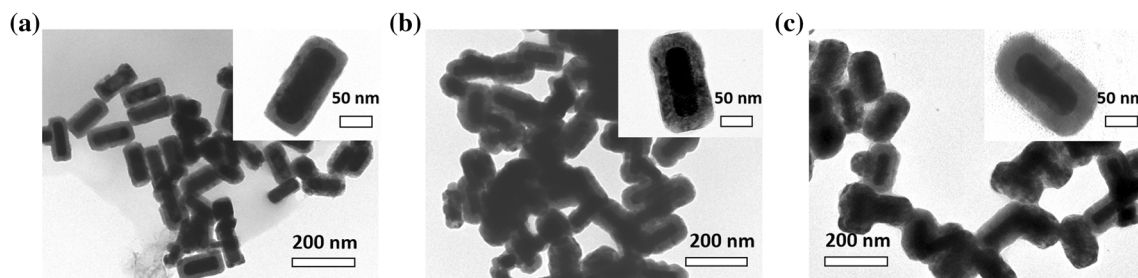


Figure 2 Au@Cu₂O nanocomposite samples with varying Cu₂O shell thicknesses. TEM images of **a** AC-35; **b** AC-65; **c** AC-100, which were produced by changing the volume of CuCl₂·2H₂O (0.05 M) from 35 μ L to 65 μ L, and 100 μ L, respectively. Insets:

enlarged pictures of the single Au@Cu₂O particle. The average lengths/widths of AC-35, AC-65, AC-100 were 143 ± 10 nm/ 67 ± 6 nm, 153 ± 13 nm/ 81 ± 11 nm, 175 ± 12 nm/ 98 ± 8 nm.

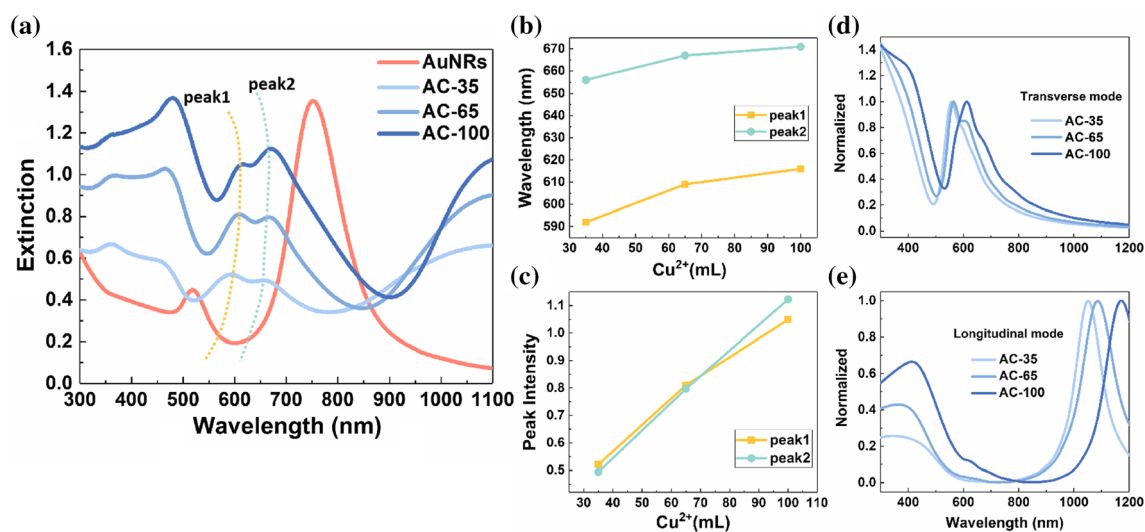


Figure 3 Extinction spectra of Au@Cu₂O NPs obtained in experiment and simulation. **a** UV–vis extinction spectra of the AuNRs and Au@Cu₂O NPs synthesized with different amounts of 0.05 M Cu²⁺ (35 μ L, 65 μ L, 100 μ L, respectively). **b** Plots of the peak wavelength with Cu²⁺ concentration. **c** Plots of the peak

intensity with Cu²⁺ concentration. **d** FDTD simulated extinction spectra of Au@Cu₂O NPs with different shell thicknesses in transverse modes. **e** FDTD simulated extinction spectra of Au@Cu₂O NPs with different shell thicknesses in longitudinal modes.

the intensity of peak1 weakened and the intensity of peak2 strengthened as the volume of Cu²⁺ increases, and the intensity of both two peaks were similar when Cu²⁺ added is 65 μ L.

To verify the plasmonic properties of the Au@Cu₂O nanostructures, we simulated the absorption spectra of single nanoparticle with different Cu₂O shell thicknesses in transverse and longitudinal LSPR modes. In Fig. 3d, transverse mode of single Au@Cu₂O with different Cu₂O shell thicknesses was normalized by the peak1. We found the tendency of the two transverse peaks gradually appeared and then it became weaker with the increasing of Cu₂O shell thickness. The extinction spectra in Fig. 3e, normalized by the longitudinal peak, shows the

longitudinal peak of Au@Cu₂O with increasing Cu₂O shell has a slight red-shift. Perfect narrow spectra were obtained in FDTD simulations, while the broadening and dampening of experimental LSPRs were observed. This is owing to the polydispersity of nanoparticles in water solution, but the simulation used the model of single particle. Although so, the change tends of plasmonic peaks calculated are in good agreement with the experimental data.

To further investigate the anisotropic effect on the extinction spectra, we have performed FDTD simulations of Au@Cu₂O with aspect ratio (AR) of 1, 2, 3, and 4, respectively. Figure S4 show the spectra of transverse (Figure S4a) and longitudinal (Figure S4b) LSPR mode of Au@Cu₂O with different AR of

AuNRs. The shell thickness of Cu_2O was set to be 20 nm in all simulations. For the case of $\text{AR} = 1$ (spheres), only one transverse peak appeared. With the increase of AR, the transverse mode became broaden and two obvious peaks appeared. For the longitudinal mode of $\text{AR} = 1$ (spheres), limited red-shift was observed. With the increase of AR (more anisotropic), the peak quickly red-shifted to near-infrared region ($\text{AR} = 4$, ~ 1300 nm). The simulated extinction spectra of Au@ Cu_2O nanosphere ($\text{AR} = 1$) and Au@ Cu_2O nanoparticles with AR of 2, 3, 4 reflected that anisotropic shape influence both the transverse and longitudinal LSPR modes.

MO as an azoic dye used on industrial scale has one or more azo bonds and aromatic rings, which is extremely stable under illumination or washing. It is helpful for textile or dye industries whereas its wastewater is harmful to biological ecosystems, such as carcinogenic and mutagenic. The azo dyes aqueous solution is highly colored even though at a low concentration [39]. Therefore, it is necessary to focus and try to solve this problem.

The catalytic activity of Au@ Cu_2O NPs was investigated by catalytic reduction of MO in the present of NaBH_4 . MO is orange and its characteristic peak is at 464 nm in water. As MO is gradually reduced, the solution will be faded to colorless and the intensity of the characteristic peak will descend, which can be monitored by the UV–vis absorption spectra. MO can be reduced by NaBH_4 , but this process is extremely slow. The time-dependent UV–vis absorption spectra were displayed in Figure S5a, in which the intensity of the characteristic peak dropped slowly and it is difficult to react completely. Then, nanoparticles, including AuNRs, AC-35, AC-65, and AC-100 were used as catalysts to reduce MO. Figure S5b and Figs. 4a–c recorded the temporal evolution of MO adsorption spectra using these particles as catalysts, and the total time of the reduction reaction respectively were 24 min, 13 min, 9 min, and 10 min. The rate of reaction using AuNRs was still low, while the rate using Au@ Cu_2O NPs had a significant improvement. Usually, we predict the reaction time shortens as the shell thickness increases. More added Cu^{2+} means a larger contact area, because we synthesized Au@ Cu_2O NPs using an equal number of Au as core. However, AC-65 reduced MO in the shortest time, which doesn't own the thickest shell. Hence, we infer a too thick shell is a disadvantage to catalyzing.

Therefore, we further tested the reaction time using pure Cu_2O NPs as catalysts in comparison with Au@ Cu_2O NPs, the time-dependent UV–vis adsorption spectra are shown in Fig. 4d. The TEM image and extinction spectra of Cu_2O NPs are displayed in Figure S6. The pseudo-first-order kinetics equation was applied to compare the rate of reaction. In Fig. 4e, the equation $-\ln(C/C_0) = kt$, was used to calculate and plot the rate constant according to Fig. 4 and Figure S5, where C is the MO concentration at reaction time t , C_0 is the initial MO concentration and k is the rate constant. Apparently, the most active catalyst is AC-65, and then is AC-100, AC-35. Pure Cu_2O particles have a better catalytic property than pure Au one, which means Cu_2O plays a more crucial role in this hybrid nanostructure to catalyze MO reduction reaction. To avoid the effect of NaBH_4 and make the rules universal, we repeated these experiments more than 3 times, and the catalytic rate and conversion with or without catalysts were exhibited in Figure S7 and Fig. 4f. Specifically, the rate of AC-65 is ~ 22 times than that of the AuNRs and ~ 4 times that of the Cu_2O NPs. The catalytic efficiency is 124% or 32% higher than the AC-35 or AC-100. The MO was completely converted using three kinds of Au@ Cu_2O NPs. The conversion of Cu_2O was approaching 90%, while the conversion without catalyst and with AuNRs only was 39% and 51%, respectively. We found that single-component nanostructures of AuNRs and Cu_2O do not perform good catalytic activities. Hence, it is speculated that the perfect catalytic activity of Au@ Cu_2O NPs is attributed to the synergistic effect of Au and Cu_2O . Within a certain thickness, the thicker shell has the better catalytic property. Once out of the range, the performance of Au@ Cu_2O nanoparticles is similar to the pure Cu_2O particles, making the catalytic properties weak. It is noteworthy that metal–semiconductor nanostructure is a series of excellent catalysts for dye degradation, and our Au@ Cu_2O nanostructures show excellent catalytic performances. Zhou's group [12] have synthesized AuNRs coating with the mesoporous SnO_2 shell to catalytic reduce 4-nitrophenol, and their catalytic activity are 3–5 times faster than that of the pure AuNRs. Gao et al. [40] have used $\text{Fe}_3\text{O}_4/\text{Au}$ NPs to degrade MO assisted by H_2O_2 as oxidant. The rate of reduction reaction enhanced 7 times compared to the $\text{Fe}_3\text{O}_4 + \text{Au}$ NPs mixture as catalysts.

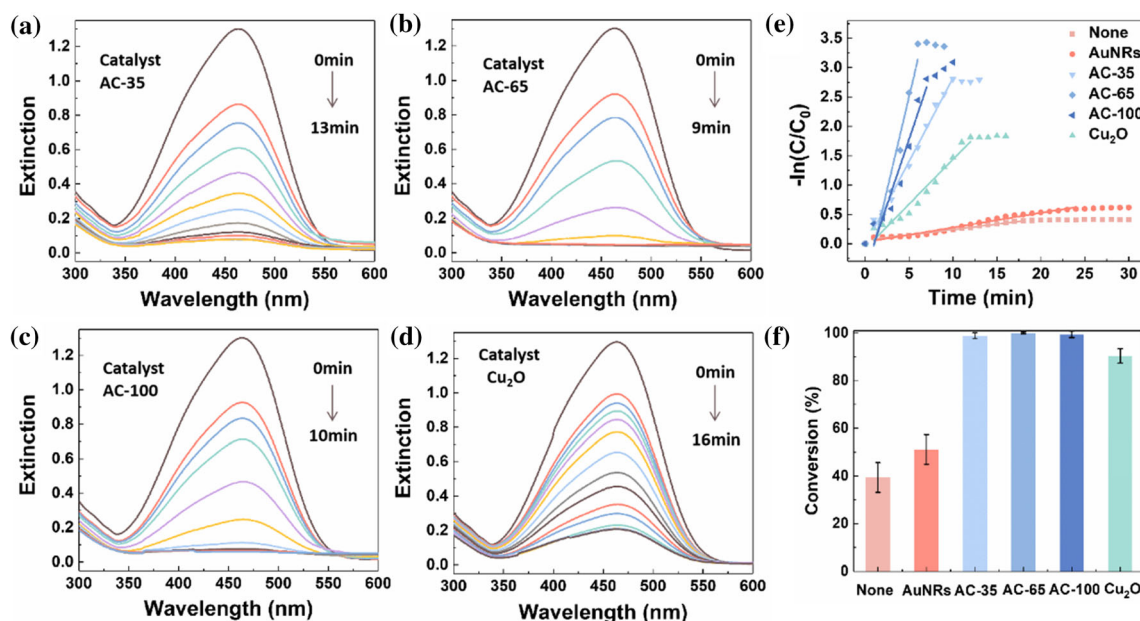


Figure 4 Time-dependent UV-vis absorption spectra of the reduction of MO by NaBH₄ using different nanoparticles as catalysts: **a** AC-35: Au@Cu₂O NPs with 15 nm Cu₂O shell; **b** AC-65: Au@Cu₂O NPs with 21 nm Cu₂O shell; **c** AC-100:

Au@Cu₂O NPs with 30 nm Cu₂O shell; **d** Cu₂O NPs. **e** Kinetic linear fitting curves for the reduction of MO of different catalysts (no catalyst; AuNRs; Au@Cu₂O NPs with three different sizes; Cu₂O). **f** Conversion for the reduction of MO of different catalysts.

A mechanism of catalytic activity on Au@Cu₂O NPs was proposed, the energy band diagram, and catalytic reduction process were illustrated in Fig. 5. It is the Schottky barrier that is helpful for electron-hole separation to render Au@Cu₂O NPs having the outstanding catalytic effects. When metal and semiconductor are in contact, the Schottky barrier may be constructed, depending on the work function of the metal and semiconductor. The Schottky barrier can

be formed only satisfying a prerequisite that: the work function of the semiconductor is significantly larger than the work function of metal, which allows for the electron to spill out from metal to semiconductor. The work function of Au is 5.1 eV. Owing to the *p*-type Cu₂O, the work function of Cu₂O depends on its crystal facets [41]. The Fermi level of Cu₂O (110) almost has no difference with Au. Therefore, a low or no Schottky barrier for holes forms between

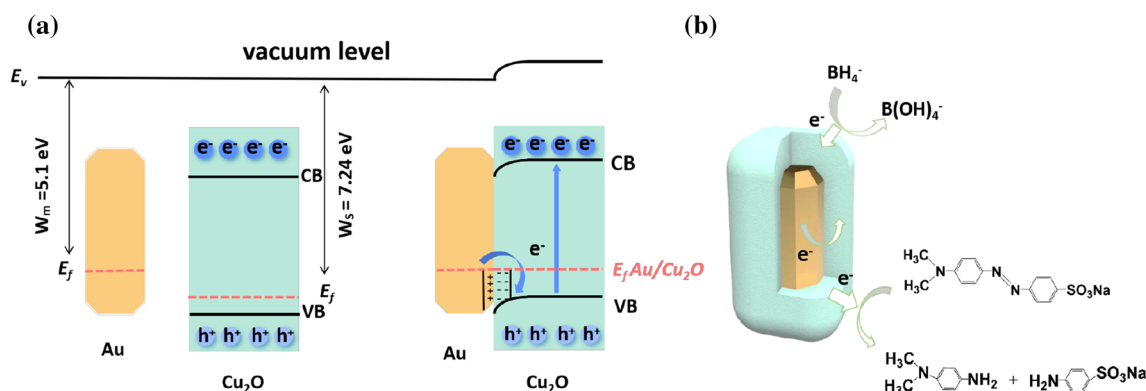


Figure 5 Schematics for mechanisms of Au@Cu₂O catalysis. **a** Energy band diagram between Au and Cu₂O of Au@Cu₂O. (W_m , W_s respectively represents the work function of Au and Cu₂O, E_v is zero potential energy in vacuum, E_f is Fermi energy

level, CB is the conduction band, and VB is the valence band.) **b** The catalytic reduction theory for MO over the Au@Cu₂O particle.

Cu₂O (110) and Au [42]. But the work function of Cu₂O (200) is 7.2 eV, suggesting that electrons prefer to transfer from Au to Cu₂O {200} facets, which mainly improves the catalytic activities of Au@Cu₂O NPs [41]. When a Schottky junction is formed, it expedites the migration of electrons. The energy band of Au and Cu₂O is curved. The Fermi level of Au goes down, and the Fermi level of Cu₂O raises up, until they are equal and an inner electric field between Au and Cu₂O is established. The Schottky junction construction at the interface can prevent the electrons from transferring back to Au, which is beneficial to restrain the recombination of e[−] – h⁺ pairs and further boost the catalytic ability. NaBH₄ is attached to the valence band of Cu₂O and supplies electrons to drive the catalytic reaction. At the same time, electrons in Cu₂O transfer from the valence band to the conduction band, and the electrons gathered at the CB of Cu₂O can capture and reduce MO, as illustrated in Fig. 5b. It is Au@Cu₂O particles that offer a platform to attach BH₄[−] ions and MO molecules and synergistically accelerate the charge migration from Au to Cu₂O for finally transferring from NaBH₄ to MO.

Summary and conclusion

In summary, we have synthesized Au@Cu₂O NPs by a simple method with binary surfactant mixtures at lower temperatures, leading to good coverage and crystalline of Cu₂O on AuNR. The synthesized Au@Cu₂O NPs are core-shell structures and the interface between the Au core and Cu₂O shell was found to be epitaxial. The size control of Au@Cu₂O was achieved by changing the volume of CuCl₂·2H₂O precursor. The prepared Au@Cu₂O NPs exhibited two transverse LSPR modes, and their intensity are related to the thickness of Cu₂O shells. In addition, the catalytic performance of all the obtained Au@Cu₂O NPs was explored by reducing MO with the aid of NaBH₄. AC-65 shows better catalytic efficiency than AC-35 and AC-100, which means the synergistic effect improves the properties of catalysts but too thick Cu₂O shells weaken the catalytic properties of Au@Cu₂O composites. Compared with AuNRs and Cu₂O NPs, Au@Cu₂O composites significantly promoted the reaction rate due to the Schottky barrier formed at the interface of Au and Cu₂O can effectively prevent the reconstruction of

electron-hole pairs. Therefore, the Au@Cu₂O nanostructures we produced which Cu₂O shells could be controlled have great potential for future catalytic fields.

Supplementary information

See the supplementary material for TEM images and extinction spectra of AuNRs and Cu₂O nanoparticles, Additional TEM, HRTEM and HAADF-STEM images of Au@Cu₂O, FDTD simulation of Au@Cu₂O NPs with different AR of AuNRs, the catalytic process of MO by NaBH₄ and AuNRs, the reaction rate of different catalysts.

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Author contributions

Y-QD: Investigation; Writing – original draft; Visualization. T-SD: Writing – review & editing; Funding acquisition; Supervision. QZ: Investigation. X-YZ: Resources. JL: Writing – review & editing; Resources. Z-QC: Funding acquisition.

Declarations

Conflict of interest The authors declares that they have no conflict of interest.

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Supporting Information

A study of size-controlled Au@Cu₂O nanocomposite for highly improved methyl orange catalytic performances

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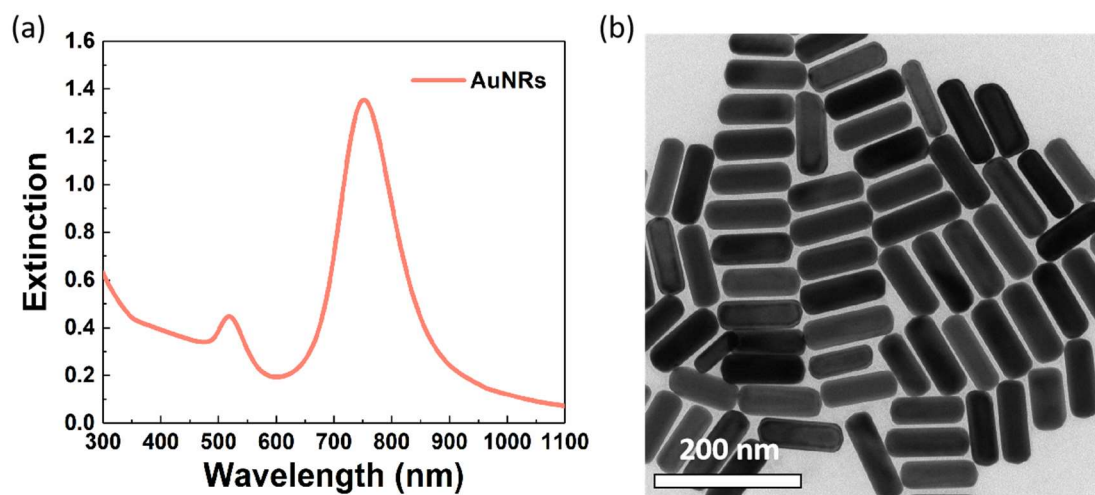


Figure S1. Extinction spectra (a) and TEM image (b) of AuNRs, having the longitudinal LSPR peaks at 752 nm. Their average lengths of AuNRs are 112 ± 9 nm, and widths are 38 ± 2 nm.

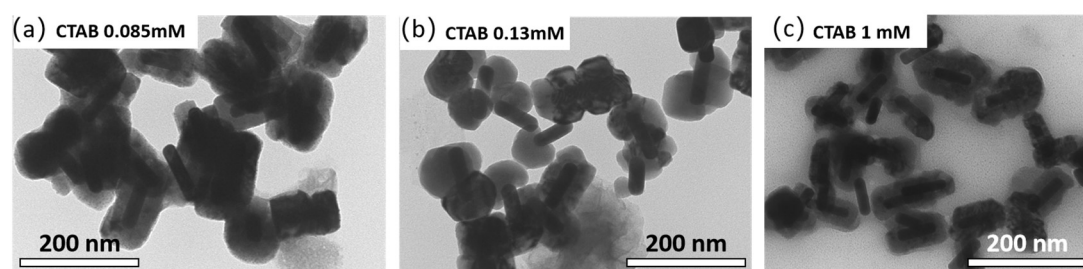


Figure S2. TEM images of Au@Cu₂O NPs, synthesized with different concentrations of CTAB.

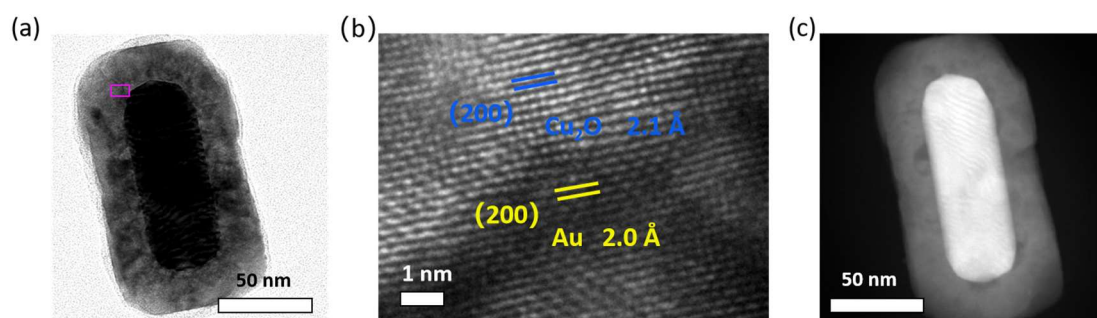


Figure S3. (a) HRTEM images of single AC-65 particle (the Au@Cu₂O with Cu₂O shell of 21 nm). (b) Interfacial HRTEM images (red square regions) on sides of AC-65 in (a). (c) The HAADF-STEM image of a single AC-65 particle.

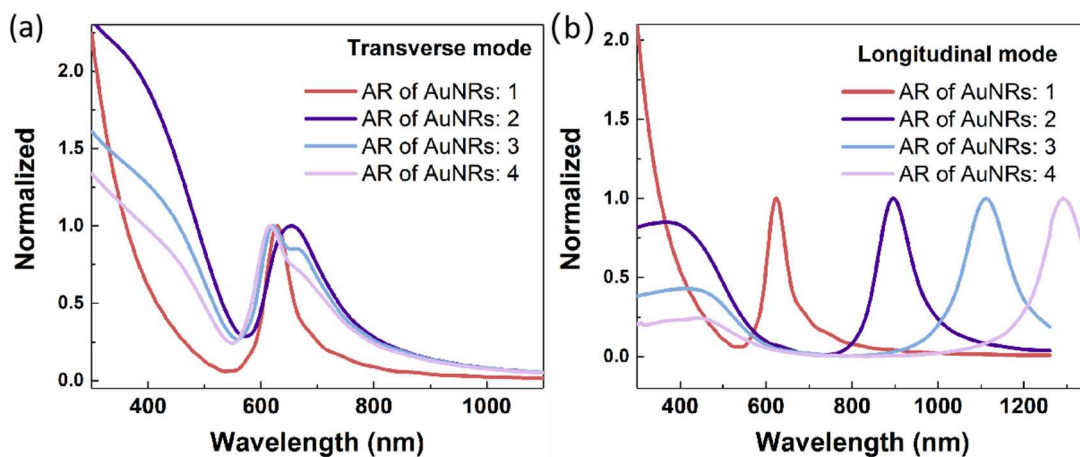


Figure S4. (a) FDTD simulated extinction spectra of Au@Cu₂O NPs with different AR of AuNRs in transverse mode. The spectra were normalized by the peak1. (b) FDTD simulated extinction spectra of Au@Cu₂O NPs with different AR in longitudinal mode. The spectra were normalized by the longitude peak. The length of AuNRs is changed at a fixed width of AuNRs, so that the AR of AuNRs in simulations are 1, 2, 3, 4. And the thickness of Cu₂O shell is set as 20 nm, as same as the Au@Cu₂O-65. The morphology of nanostructure that the AR=1 of AuNRs is namely Au@Cu₂O nanosphere.

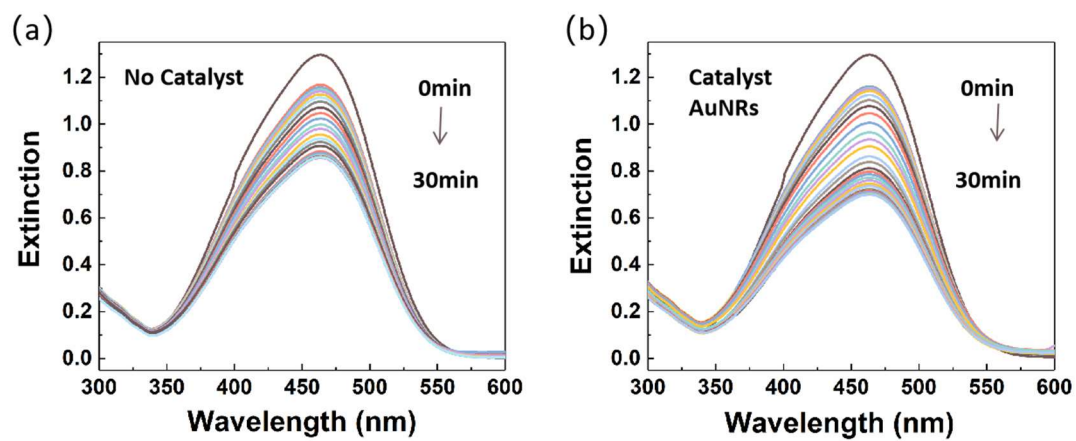


Figure S5. Time-dependent UV-vis absorption spectra of the reduction of MO by NaBH₄ using: (a) No catalyst; (b) AuNRs.

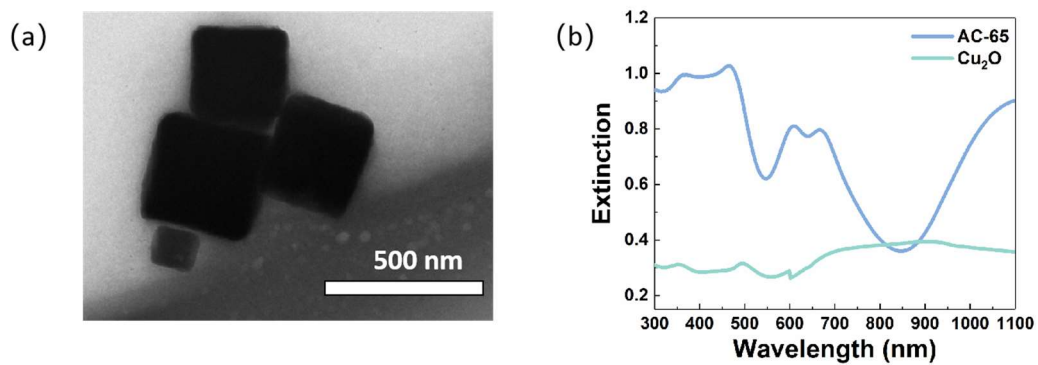


Figure S6. (a) TEM image of Cu₂O NPs; (b) UV-vis absorption spectra of Au@Cu₂O NPs and Cu₂O NPs. Two kinds of nanoparticles were synthesized by addition of 65 μ L of 0.05 mM CuCl₂·2H₂O

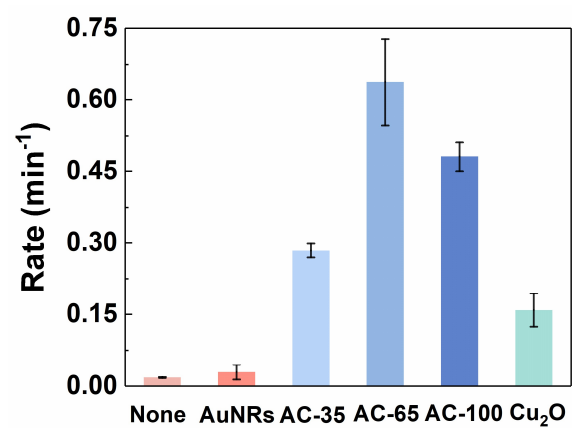


Figure S7. Comparison between the reaction rate of different catalysts: no catalyst, AuNRs, Au@Cu₂O NPs with three different sizes, and Cu₂O NPs.